

Nitrogen Doped Carbon Nanofibers Derived from Water-Soluble Precursors

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Abstract. Nitrogen doped carbon fibers were synthesized from polyvinyl alcohol (PVA), a water soluble precursor. The PVA fine fibers were first developed by centrifugally spinning an aqueous based solution using Forcespinning® technology. The precursor fibers were then exposed to sulfuric acid vapors to partially carbonize and stabilize the fibers for further heat treatment. For nitrogen doping, the fibers were exposed to two different heat treatment routes. One was under a nitrogen atmosphere at 850°C followed by exposure to ammonia gas at 500°C. The second route consisted of heating the treated fibers in pure ammonia gas only, up to 850°C. Both heating schemes resulted in carbon based fibers that showed evidence of nitrogen content as shown by energy-dispersive X-ray spectroscopy. The second route showed an effective doping of the carbon fiber with nitrogen atoms, measured by X-ray photoelectron spectroscopy, which indicated that the nitrogen atoms were fully incorporated into the carbon framework.

Keywords: polyvinyl alcohol, carbon nanofiber, Forcespinning®.

1 Introduction

Carbon-based materials have many applications in a wide range of areas such as electrode components (energy conversion and storage applications), filtration, structural-reinforcement, oil absorption, and electromagnetic shielding to mention some [1]. Carbon fibers, in particular have been extensively used in the aerospace, sports, and automobile industries. It is estimated that the global carbon fiber market has grown to over 12,500 tons per year, indicating that carbon fiber has become not only of great technological importance but a strong commercial entity. Carbon fibers also have the potential to undergo chemical modification or doping to obtain novel material properties. These fibers may find potential applications in smart textiles such as in thermoelectric fabrics, where materials could produce electricity by harvesting heat generated by the body [2]. Such a technology would provide a new way for developing novel power generating clothing and self-powered portable electronics. Durable and flexible thermoelectric felts, incorporating doped carbon-based materials, are a promising alternative to traditional thermoelectric alloy based materials [3-5]. Most commercially available thermoelectric materials consist of inorganic bismuth telluride based alloys which operate at portable device temperatures (e.g. below 200°C) and possess an efficiency or figure of merit ZT close to one [6, 7]. A high figure of merit (ZT) is desirable and can be produced by increasing the Seebeck coefficient (α) and electrical conductivity while keeping a low thermal conductivity (k). Recent studies have shown and predicted improvement in the thermoelectric output by the doping of carbon nanotubes (CNTs) [8]. The improved Seebeck effect by impurity doping of CNTs has shown the possibility to make organic thermoelectric devices [9, 10] in addition to nanotransistors, sensors, and diodes.

A few studies have shown the effects of doping carbon fibers, for example, Choi *et al.* studied the doping of fibers to develop supercapacitor electrodes [1] in which the pyrolysis of a nitrogen containing polymer was electrochemically grown on carbon fiber paper. Other studies have developed doped carbon

fibers through electrocatalysts in metal-air batteries [11] where polystyrene and polyacrylonitrile underwent pyrolysis to form doped carbon fiber. Previous work has shown the doping of CNTs by either arc growth methods whereby nitrogen is introduced during pyrolysis or by chemical modification after synthesis [3, 9]. Results of these studies have indicated that nitrogen doping of CNTs results in a negative Seebeck due to strong donor states [12]. The n-type or electron carrier behavior is an important component of thermoelectric modules where hetero-junctions improve the overall efficiency. In this study we demonstrate the production of an “n”-type material by nitrogen doping of the carbon fiber produced by pyrolytic routes, which results in the introduction of nitrogen to the carbon backbone.

A traditional and common method for producing carbon fibers is based on oxidation, carbonization, and graphitization of polyacrylonitrile (PAN) fibers. PAN has been the precursor of choice due to its high-tensile strength, though the high cost associated with the low yield, energy intensive oxidation stabilization steps, and high carbonization temperature, has hindered its wide acceptance as the material of choice for many applications. The usage of PAN based carbon fibers has therefore been limited to high performance applications. During the carbon conversion process, about 50% of PAN mass is lost and its volatiles require abatement methods to control the toxic species. The costs associated with post-processing techniques such as stabilization, sizing, surface treatment, and material handling are low compared to the cost of the raw material (PAN). Recent attempts have been made to use alternative lower cost precursors such as meso-phase pitch based carbon fiber [13]. Alternate precursors such as PVA are ideal for carbon fiber production due to its low toxicity, cost effectiveness, and sufficient carbon content (54 % carbon by mass) [14, 15], and provide the sufficient mechanical strength for applications in filtration and electrodes.

In the case of carbon fibers, there are several methods available to produce fine fibers (nano to single digit micron) such as: wet chemistry methods [16], solution and melt blowing, and spinning methods (wet, dry and melt spinning). Wet chemistry methods still have low nanofiber (NF) yield and significant scalability issues. Solution and melt blown [17, 18] processes have been proven reliable and cost effective for micron size fibers, although for NFs, the process consumes large amount of heated gas and is not feasible for industrial scale up. Fibers produced through electrospinning [19-21] can range from tens of nanometers to a few micrometers, however at the lab scale, the outcome is low, averaging 0.01-1.0 gram per hour. Other methods, such as “Island-in-the-Sea” bi-component fiber spinning, combined with thermo-chemical stabilization and carbonization, have shown to produce continuous and unidirectional carbon NFs with various shapes [22]. However, the “fugitive” polymer may require removal by an organic solvent at a specified temperature, and therefore costly if extended to mass scale production.

The Forcespinning® (FS) method has shown capability to produce fine fibers from melt and solution spinning through centrifugal spinning. Since it does not require electric fields, it broadens the choice of materials to be spun into NFs [21, 23, 24]. The process at the industrial scale has shown production rates of up to hundreds of meters per minute depending on desired basis weight, which is highly controllable in the process. Herein, nitrogen doped carbon fibers were mass produced by converting the PVA fine fibers to carbon NFs through a dehydration process, induced by exposure to acid vapor up to 200°C described by Cremer *et al.* [25-27], followed by carbonization at low temperatures ($T < 800^{\circ}\text{C}$) in an inert gas or in pure ammonia gas.

2 Materials and Methods

2.1 Materials

Polyvinyl alcohol (Average M_w 85-124K, 96% hydrolyzed) and 98% sulfuric acid were purchased from Sigma Aldrich. Pure anhydrous ammonia gas was purchased from Airgas. Fibers were produced utilizing the lab scale Cyclone™ L-1000M from FibeRio Technology Corporation.

2.2 Sample Preparation

The nonwoven nitrogen doped carbon fiber mat was prepared as follows: A 10% (wt./wt.) solution of PVA was prepared by adding and dissolving 1.11 g of PVA to 10 mL of deionized water and stirred for 4-6 hours in an oil bath at 75 °C. The nonwoven mat was prepared as follows: 1 mL of the obtained

PVA aqueous solution was injected into the Cyclone spinneret and spun at 7000 RPM for 1 minute. The fibers were allowed to collect on a substrate attached to a vacuum deposition system to form a nonwoven mat. This procedure was repeated until the mat was approximately 35 grams per square meter. The mat was treated with 98% sulfuric acid vapors in a heated semi-closed apparatus. Approximately 1-5 mL of sulfuric acid was placed in a beaker, and the nonwoven mat was sandwiched between the brim of the beaker and a glass funnel (or plate). The nonwoven sulfuric acid treated PVA fiber mat was washed in water until the acid residue was completely removed and then dried in air. Pre-oxidation was carried out at up to 200 °C in air. The nonwoven mat was then placed in a tube furnace and carbonized in nitrogen or pure ammonia gas, with a ramping rate between 2-10 °C min⁻¹, and held at the final temperature for less than 30 minutes.

2.3 Characterization

The morphology of the fiber mats was extensively studied by scanning electron microscopy (SEM, Sigma VP, Carl Zeiss, Germany). Elemental quantification of the developed fiber mats was studied with energy-dispersive X-ray spectroscopy (EDAX) (Model Octane Super within the SEM). The extent of doping of the carbon fibers with nitrogen was determined by X-ray photoelectron spectroscopy (XPS) with a ThermoScientific K- α XPS surface analysis instrument with a micro-focused monochromated Al K- α X-ray source.

3 Results and Discussion

3.1 Nitrogen Doping of Carbon Fiber

Two heat-treating schemes were explored to determine the extent of nitrogen doping on the acid vapor treated mats. For the first method or route (Fig. 1), partial carbonization of the PVA nonwoven mat was carried out through the acid vapor treatment followed by washing of the mats to reduce acid traces [25]. The mats were fully carbonized when heat treated under nitrogen at 850 °C. The temperature was then decreased to 500 °C, and the mat was exposed to pure ammonia gas for one hour and then cooled to room temperature.

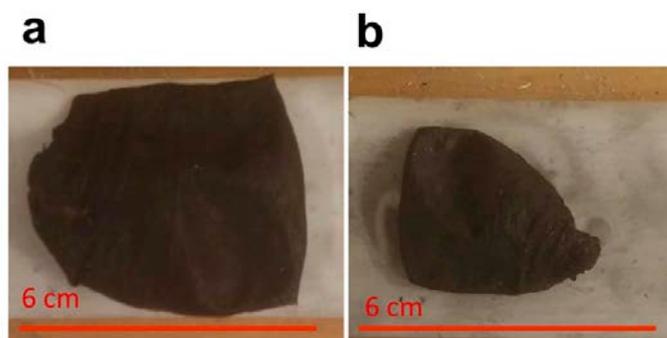


Figure 1. Nitrogen doped fibers - route 1: Carbonization procedure of the (a) sulfuric acid vapor treated polyvinyl alcohol nonwoven mat was carried out at 850°C followed by (b) ammonia treatment at 500 °C.

In the second heating scheme, partial carbonization of the PVA nonwoven mat was also carried out with the acid vapor treatment followed by washing of the mats to reduce acid traces. However, the pyrolysis route involved heating the acid vapor treated mats under pure ammonia gas, from room temperature up to 850 °C, followed by cooling to room temperature. Results are shown in Fig. 2.

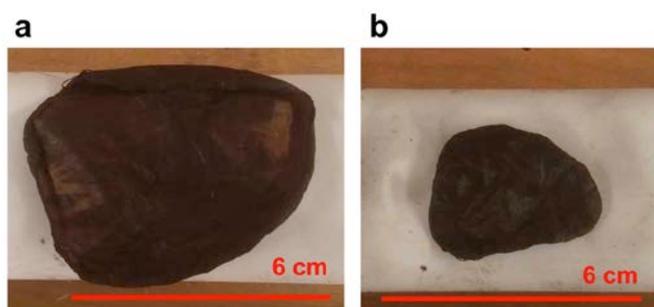


Figure 2. Nitrogen doped fibers route - 2: The PVA mat was first acid vapor treated (a) and then heat treated (b) in pure ammonia gas up to 850 °C at a ramping rate of 4°C min⁻¹.

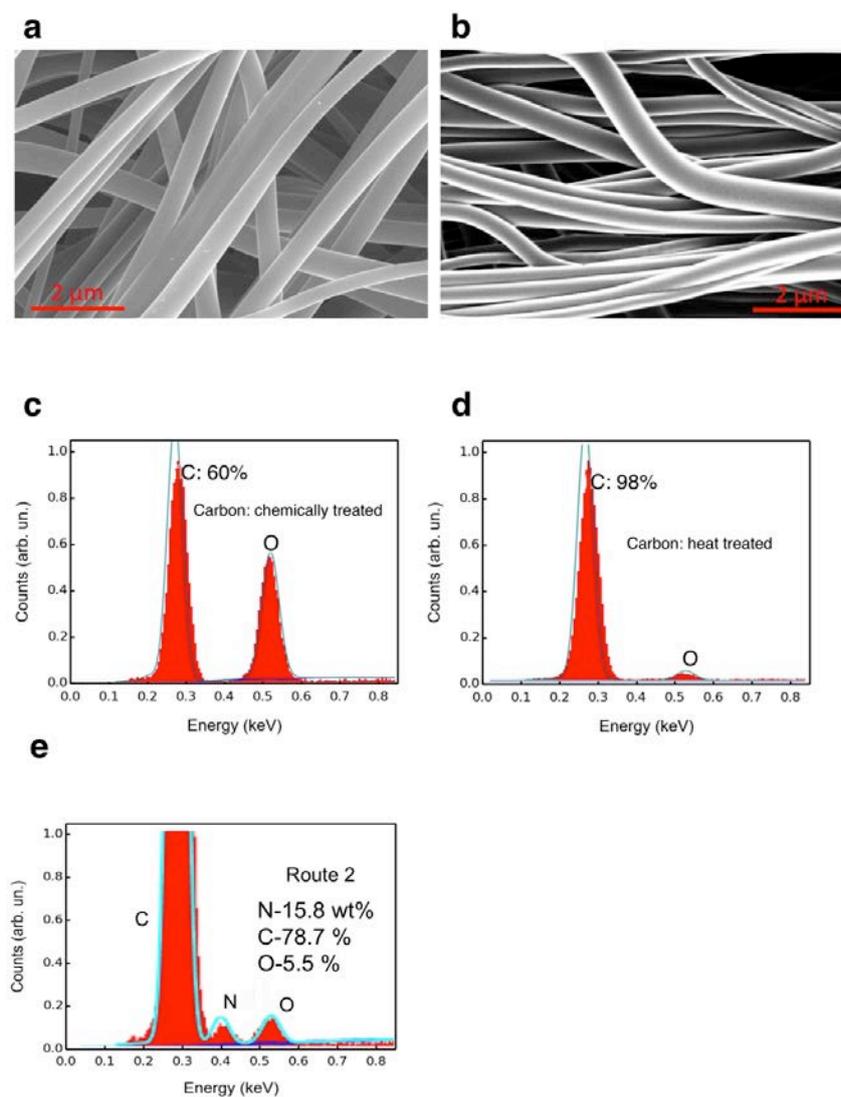


Figure 3. SEM and EDAX spectroscopy of chemically and heat treated fibers for routes 1 and 2. Scanning electron microscope images are shown for the nitrogen doped fibers through route 1 (a) and 2 (b). EDAX spectra show evidence of carbon and oxygen atoms for (c) the sulfuric acid vapor treated PVA fibers. When the fibers are heat-treated the peak intensity of oxygen is reduced (d). When the carbon fibers are heat treated under ammonia, via route 2, the nitrogen content increases to 15.8 wt.% nitrogen.

3.2 Changes in Carbon Fiber Properties

The chemical composition of the carbonized fibers for route 1 was analyzed using energy-dispersive X-ray spectroscopy (EDAX). EDAX and SEM images of the fiber mats are also shown for route 2, during various stages of the carbon formation process (Fig. 3). There are basically two peaks representing carbon and oxygen (Fig. 3c), and the intensity of the oxygen peak is reduced significantly when the mat undergoes heat treatment in nitrogen (Fig. 3d).

Fig. 4. depicts an X-ray diffraction pattern of the nitrogen-doped carbon fibers following route 1. During acid treatment the crystalline structure of the fibers disappear and there is peak broadening and upshift to a higher angle centered between $2\theta = 20\text{-}30^\circ$. The acid and heat treated fibers show a broad (002) peak which corresponds to the disordered micrographitic stacking. Heat treatments up to 850°C show a broad (10) peak between $2\theta = 40\text{-}50^\circ$ corresponding to the graphite plane [25, 26]. The crystal spectra indicate carbonization during acid vapor and subsequent heat treatment (24°), whereas the PVA precursor fibers have a crystalline peak at $2\theta = 19^\circ$. The ammonia treated carbon fibers, following route 1, show a broadening of the (002) graphite plane. Similar reports for carbon fiber studies have shown amorphous peaks centered around $20\text{-}30^\circ$ [28].

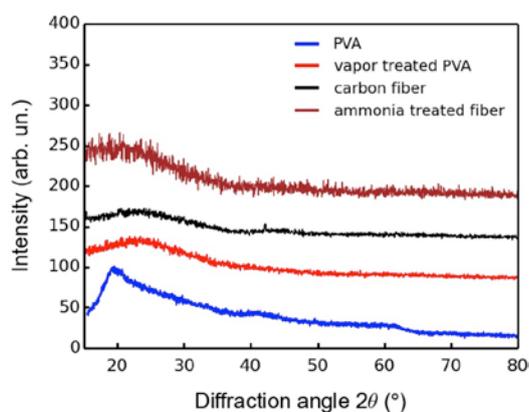


Figure 4. X-Ray Diffraction patterns of PVA, acid treated PVA, carbon fiber, and nitrogen doped carbon fiber (route 1). The precursor PVA nanofibers exhibit a semi-crystalline structure with a narrow peak at $2\theta = 19^\circ$. During acid treatment the crystalline structure of the fibers disappears, and there is peak broadening and upshift to a higher angle centered between $2\theta = 20\text{-}30^\circ$. The acid and heat treated fibers show a broad (002) peak which corresponds to the disordered micrographitic stacking. Heat treatments up to 850°C show a broad (10) peak between $2\theta = 40\text{-}50^\circ$ corresponding to the graphite plane. The ammonia treated carbon fibers show a broadening of the (002) peak.

Fig. 5 and 6 show XPS spectra for the carbon fibers, produced by route 1 and 2, that were treated with ammonia. The survey spectrum (Fig. 5(a) and 6(a)), indicates the presence of carbon, nitrogen, and oxygen on both samples which is consistent with the EDAX spectra. For the sample produce by route 1 (Fig. 5(a)), the C 1s, N 1s and O 1s peaks appear at binding energies of 284.5, 399.6 and 532.8 eV, with a weight percent concentration of 93.3 wt.% C, 5.1 wt.% O, and 1.6 wt.% N, respectively. Fig. 5b, shows the C 1s graphitization peak at 1225.1 eV, along with an inset schematic showing nitrogen incorporation, which is in agreement with the XRD data that indicate formation of the (002) graphitic plane between $20\text{-}27^\circ$. Several reports have indicated that nitrogen states between 400.5-401.5 eV can be associated as quaternary or graphite-like nitrogen, which is incorporated into the structure of the extended aromatic π -system in carbon nanofiber [29-34]. The assignment of chemical groups from XPS spectra can be achieved through peak deconvolution obtained through curve fitting. The decomposed spectra for the N 1s state show a graphitic-like nitrogen peak centered at 400.6 eV (Fig. 5(b, d)) which corresponds to a N atom bonded to three sp^2 carbons.

Fig. 5(c) shows the deconvoluted spectra for seven C 1s bonding states with the corresponding binding energies: C_I -284.5, C_{II} -285.7, C_{III} -286.5, C_{IV} - 287.4, C_V - 288.5, C_{VI} - 289.8, and C_{VII} - 291.2 eV. The main C-C bonding peak centered at 284.5 eV (C_I) has the greatest intensity, and corresponds to sp^2

bonding found in graphitic structures, while the peak at 285.7 eV (C_{II}) can be assigned to the C-C sp^3 bonding state. Previous reports by Ismagilov *et al.* have indicated that nitrogen doping of carbon fiber results in the C 1s peak shifting to higher binding energies from 284.3 eV (undoped highly oriented pyrolytic carbon fiber) to 284.6 eV (8.2 wt.% for nitrogen doped) [29]. The shift is due to the disordering of the graphite-like structure after incorporation of nitrogen. Undoped carbon fiber produced herein has a C_I peak at 284.6 eV, which is greater than the reported C 1s peak for highly oriented pyrolytic carbon fiber. With respect to the latter, graphite in carbon nanofiber is less ordered and has been reported to have a C 1s graphitic peak at 284.5 eV, and may not show a C 1s peak shift when treated in an ammonia mixture [35]. The remaining five C 1s bonding states can be assigned to the bonding of carbon to oxygen on the surface in the form of various functional groups [36-38]. The peak found at 286.5 eV (C_{III}) can be assigned to C-O-R (R= C, H), which can either be an epoxy (C-O), ether (C-O-C), or hydroxyl (C-OH) group. The peak found at 287.4 eV (C_{IV}) corresponds to the carbonyl (C=O) found on the aldehyde/ketone groups. A carboxyl/carboxylate group (O=C-OH / O=C-O⁻) can be assigned to the peak at 288.5 eV (C_V) and 289.8 eV (C_{VI}). The peak with the highest binding energy for the C 1s state at 291.2 eV (C_{VII}) can be assigned to the $\pi-\pi^*$ “shake-up” satellite, which accompanies the photoionization process of the 1s electron in the conjugated π -system found in graphite [29].

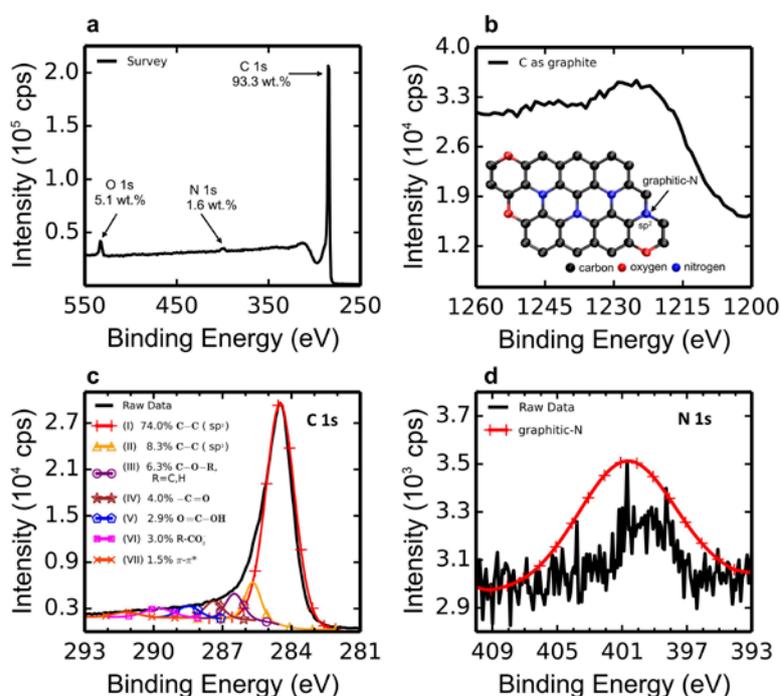


Figure 5. XPS of nitrogen doped carbon fiber - route 1. (a) The survey indicates the presence of carbon, oxygen, and nitrogen with a peak at 400.6 eV. (b) The peak at 1225.1 eV indicates the presence of graphite (inset) and the decomposed C 1s states shown in (c) indicates a principal peak at 284.5 eV corresponding to the sp^2 hybridized C found in graphite. The deconvoluted N 1s state shows a bonding state at 400.6 eV, which can be assigned to a quaternary or graphitic type N found in graphite.

The XPS spectra of the sample produced by route 2 (Fig. 6a) show the C 1s, N 1s and O 1s peaks at binding energies of 286.1, 398.1 and 532.1 eV, with a weight percent concentration of 78 wt.% C, 14.3 wt.% O, and 8.0 wt.% N, respectively. Fig. 6(c) shows the deconvoluted spectra lines for the C 1s state where five bonding states were identified with the corresponding binding energies of 284.6 (C_I), 286.3 (C_{II}), 287.8 (C_{III}), 290.2 (C_{IV}), and 291.7 (C_V) eV.

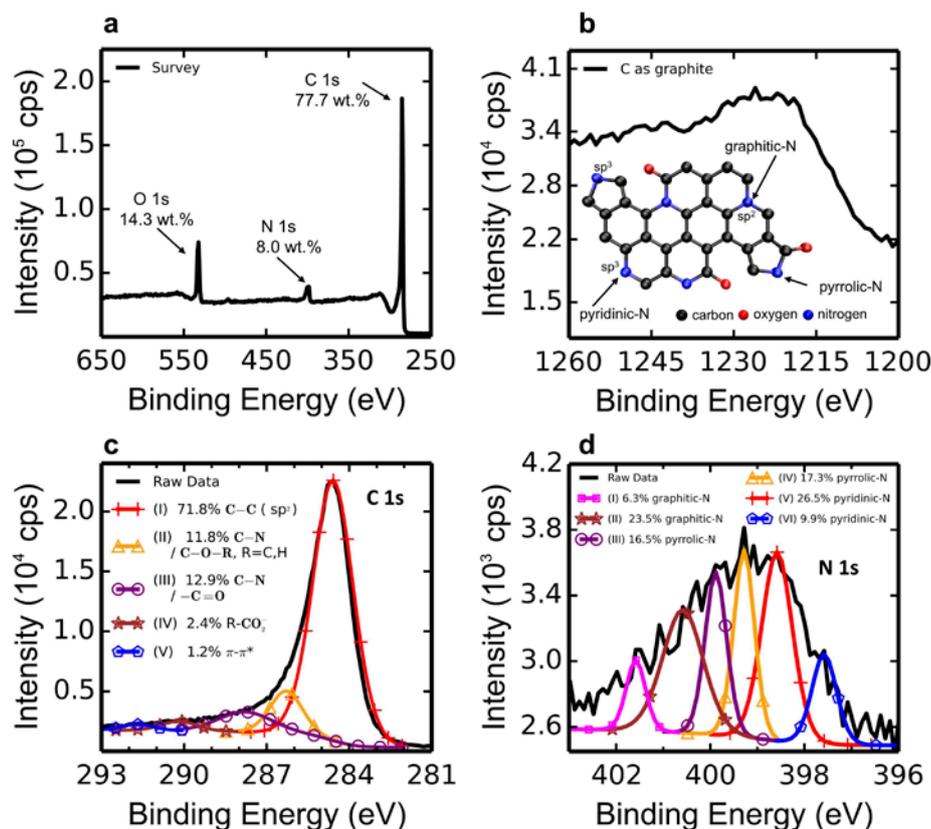


Figure 6. XPS of nitrogen doped carbon fiber - route 2. (a) The survey also indicates the presence of carbon, oxygen, and a nitrogen peak at 398.1 eV. (b) The peak at 1226.1 eV indicates the presence of graphite. The inset shows a schematic of nitrogen incorporated into the graphite structure. (c) A detailed analysis of the C 1s peak indicates five 1s bonding states: a principal sp^2 hybridized C-C bond along with other carbon/nitrogen type functional groups. (d) The deconvoluted spectrum for The N 1s peak shows that the carbon framework has nitrogen incorporated within the graphitic ring structures (graphitic: $N_{I-401.6}$ eV, $N_{II-400.6}$) in addition to other sp^3 hybridized nitrogen bonding states.

The principal C 1s binding energy peak (284.6 eV, C_{I}), for the 8 wt.% N doped carbon fiber produced by route 2, is slightly greater than reported values for undoped carbon fiber (284.3 eV). In one study, Ismagilov *et al.* showed that the C 1s peak shifted to higher binding energies from 284.3 to 284.6 eV for an 8.2 wt.% nitrogen doped carbon fiber, which was due to the increased structural disorder in the graphitic network [29]. For route 2, there are notable differences for the remaining C 1s bonding states, which can also be assigned to different bonding functionalities of carbon-oxygen on the surface [36-38]. Unlike route 1, the C-C sp^3 hybridized bonding state is not evident. The C_{II} bonding state (286.3 eV) can be assigned to a C-O-R (R=C,H) and C-N bond which can span from 285.2-288.5 eV [38]. This is evident from the increased bonding character, which is 11.8 % for the C_{II} state in route 2 compared to 6.3% for the same bonding functionality in route 1. The C_{III} state (287.8 eV) may correspond to a carbonyl ($-C=O$, found on the aldehyde/ketone groups) and a C-N bond, given that the bonding state is 12.9%, which is greater than the state corresponding to the carbonyl in route 1 at 4.0%. The C_{IV} (290.2 eV) bonding state can be assigned to a carboxylate ($O=C-O$), and the C_V peak (291.7 eV) with the highest binding energy to the $\pi - \pi^*$ "shake-up" satellite [29, 39].

Fig. 6b, shows the C 1s graphitization peak at 1226.1 eV with an inset schematic representing the different bonding configurations of nitrogen in the graphitic plane. There are generally at least five different nitrogen bonding types that have been reported which fall into a range of different binding energies [29-34]: pyridinic-N (397.3-398.8 eV, 402.1-402.5 eV), pyrrolic-N (399.5-400.3 eV), pyridone-N (400.2 eV), quaternary or graphitic-N (400.5-401.2 eV), and pyridinic oxide-N (403.5-405 eV). One

report on the nitrogen doping of carbon fiber [31], containing carbon nanoparticles, showed a pyridinic state ranging from 396-400.5 eV, a pyrrolic state from 399-402 eV, and a graphitic state from 399-403.5 eV. Deconvolution of the N 1s peak suggests that the carbon network is doped with three nitrogen types (Fig. 6d). Six N 1s bonding states were identified and denoted, with the corresponding binding energies, as: graphitic N_I-401.6, quaternary or graphitic (sp² hybridized) N_{II}-400.6, pyrrolic N_{III}-399.9, pyrrolic (sp³ hybridized) N_{IV}-399.3, pyridinic N_V-398.6, and pyridinic N_{VI}-397.6 eV, respectively [27-29]. Each of the three nitrogen bonding types (pyridinic, pyrrolic, and graphitic) shown in Fig. 6d has two peaks, which can be attributed to the bonding environment. For instance, the presence of oxygen on the surface of the graphitic network (14.3 wt.% oxygen) suggests (inset Fig. 6b) two possible types of pyrrolic or pyridinic nitrogen: an N-C-C and N-C-O type bond. Graphitic-N can be found either in-plane or along the edge. Kiuchi *et al.* reported on the nitrogen doping of pyrolytic graphite prepared by low energy nitrogen ion sputtering and subsequent annealing at less than 800 °C [40]. As the nitrogen concentration decreased, two graphitic nitrogen components were identified: a graphitic-N in the basal plane (nitrogen bonded to three carbon atoms) at 400.9 eV, and a graphitic-N in the zigzag edge and/or vacancy sites with a binding energies of 401.9 eV. The difference in the binding energy or chemical shifts is due to the chemical environment, e.g. placement of nitrogen atom at different sites or inequivalent positions. In general, the assignment of peaks and nitrogen coordination is not always straightforward [41], especially with overlapping peaks that appear within a narrow window or within the full-width-half-maximum (FWHM less than 2.5 eV). For this reason, theoretical models have often been used to assist in peak identification. Casanovas *et al.* carried out first principles calculations which suggested that the zigzag graphitic-N peak shifts to higher binding energy by approximately 1 eV compared to the in-plane graphitic-N [42]. The graphitic-N peaks (N_I-401.6 and N_{II}-400.6) reported herein follow a similar peak shift.

When the mats are nitrogen doped following route 1, the surface nitrogen content increases to approximately 1.6 wt.% (Fig. 5a). The electrical conductivity was on average $6.1 \times 10^{-3} \text{ S} \cdot \text{cm}^{-1}$ (volume resistivity of $165 \text{ } \Omega \cdot \text{cm}$) for the acid vapor treated mats, which increased to $2.46 \text{ S} \cdot \text{cm}^{-1}$ ($0.407 \text{ } \Omega \cdot \text{cm}$) for the carbon mats, and increased further to $4.6 \text{ S} \cdot \text{cm}^{-1}$ ($0.22 \text{ } \Omega \cdot \text{cm}$) for the 1.6 wt.% nitrogen doped carbon fibers. The nitrogen content for the doped carbon fibers produced from route 2 (8 wt.%), however, shows a much lower conductivity of $4.2 \times 10^{-2} \text{ S} \cdot \text{cm}^{-1}$ (volume resistivity of $24 \text{ } \Omega \cdot \text{cm}$). Ismagilov *et al.* reported on the doping of carbon nanofibers with varying concentrations of nitrogen (0-8.2 wt.%) produced by the decomposition of ethylene and an ethylene/ammonia mixture over a catalyst at 550 °C [29, 35]. In the latter study, the conductivity was $9.1 \text{ S} \cdot \text{cm}^{-1}$ for the undoped carbon fiber structure ('herringbone'), and $15.4 \text{ S} \cdot \text{cm}^{-1}$ for a 3.1 wt.% nitrogen doped carbon nanofiber. However, the conductivity decreased to $6.1 \text{ S} \cdot \text{cm}^{-1}$ for an 8.2 wt.% nitrogen doped carbon fiber sample. The decreased conductivity was due to the increased defectiveness of the graphite-like structure as shown by Raman spectroscopy (shown by the broadening of the D band and an increase in the ratio of integral intensities of lines D and G). A similar trend by Chiang *et al.*, that involved the boron doping of carbon nanotubes, also demonstrated a reduction in conductivity with increased boron content which was due to the formation of defects [43].

Decreased conductivity, due to increased graphitic defects, in the nitrogen-doped carbon fiber is connected to the particular nitrogen bonding type (e.g. sp³ hybridized pyridinic-N). As mentioned, Ismagilov *et al.* showed that 1.7 and 3.1 wt.% nitrogen doped carbon fiber preferentially contains a graphitic-N peak (400.5 eV), which has an additional electron that can delocalize to the π -system in carbon fiber. However, at higher nitrogen concentrations of 8.2 wt.%, the fraction of the pyridinic state increases (398.7 eV); the pyridinic XPS peak intensity becomes greater than the graphitic-N and a broad peak (409 eV) corresponding to the pyridinic $\pi - \sigma^*$ transition appears (excitation of unshared electron pair to vacant σ^* orbital). With respect to the latter, the increased pyridinic state therefore leads to an increase fraction of localized states or a decreased localization radius of the electron wave function (denoted as " \mathbf{a}_{loc} " where $\mathbf{a}_{\text{loc}} = 6.1 \text{ nm}$ for 1.7 wt.% N and $\mathbf{a}_{\text{loc}} = 4.9 \text{ nm}$ for 8.2 wt.% N). This results in decreased current carrier mobility or conductivity in the sample.

With respect to the route 2 syntheses herein, the observed decrease in the electrical conductivity is due to the presence of additional sp³ hybridized nitrogen states (pyridinic and pyrrolic-N) which contribute to increased localized states. This is in contrast to route 1 where only one graphitic-N state (additional electron added to π -system) was incorporated into the carbon backbone at a lower nitrogen content of 1.6 wt.%, which was shown to be more conductive. For route 2, there is a competition of two

independent processes when incorporating nitrogen into the graphite lattice. As a result, an optimum nitrogen doping concentration may exist, whereby adding an additional electron carrier to the delocalized π -system (graphitic-N) on the graphite layer, can overcome the disordering of the graphite lattice (from pyridinic or pyrrolic states).

4 Conclusion

In this study we showed the production of nitrogen doped carbon fiber nonwoven mats from a low-cost water soluble polymer precursor using the high-throughput Forcespinning[®] (FS) technology. The developed doped carbon fibers have potential use as n-type carbon materials where improvement in thermoelectric high power output is needed. The nitrogen doping method allows for the introduction of electron rich impurities into the carbon framework, thereby creating donor states, and lead to improved conductivity with respect to the un-doped carbon fibers. For industrialization of carbon-based materials, this would be an ideal production method since it is simple and effective for low-cost production, scalability, and reproducibility. In terms of processability and performance, the carbon-based material can be functionalized or modified by doping to create advance or novel materials for thermoelectric applications. The presented carbon production method is a novel and efficient way to produce carbon fiber, and the dehydration induced process, coupled with low temperature carbonization, presents an attractive method to develop other types of doped carbon nonwoven mats (e.g. p-type carbon materials). The developed carbon based materials could be used "as is" or embedded in polymer composites for reinforcement. Further investigations are underway to study the thermoelectric properties of modules containing p and n-type nonwoven multifunctional carbon fiber mats.

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